# Polyhedral Packing in Giant Cells 

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#### Abstract

The structure of alloys can be described in terms of packing of polyhedra. The present paper is a study of the packing of polyhedra in giant cells: $\mathrm{Ca}_{3} \mathrm{Ag}_{8}$ (cI44); $\mathrm{Tl}_{7} \mathrm{Sb}_{2}$ (cI54); $\mathrm{Ti}_{2} \mathrm{Ni}$ (cF96); $\mathrm{Th}_{6} \mathrm{Mn}_{23}$ (cF116); $\mathrm{Cu}_{41} \mathrm{Sn}_{11}(\mathrm{cF} 416) ; \mathrm{Sm}_{11} \mathrm{Cd}_{45}$ (cF448). $\mathrm{Ca}_{3} \mathrm{Ag}_{8}, \mathrm{Ti} \mathrm{i}_{2} \mathrm{Ni}$, and $\mathrm{Th}_{6} \mathrm{Mn}_{23}$ can be described in terms of packing of one polyhedron; $\mathrm{Tl}_{7} \mathrm{Sb}_{2}$ in terms of packing of two polyhedra; $\mathrm{Cu}_{41} \mathrm{Sn}_{11}$ and $\mathrm{Sm}_{11} \mathrm{Cd}_{45}$ in terms of packing of four polyhedra. The polyhedra model proved successful for about 50 structure types treated up to now. The coordination numbers of the individual atoms in the treated structure types determined by the "maximum gap" method range from 8 to 16 . © 1986 Academic Press, Inc.


## 1. Introduction

Consideration of coordination spheres of particular atoms has proved to be a useful approach to understanding the crystal chemistry of alloy structures. The search for a model to describe these structures led us to the polyhedral packing principle, which enables us to construct even complicated structures with few different polyhedra. The aim of this approach is to present a simple description of the complex alloy structures and to systematize the different known structure types.

Description of structures in terms of packing of single polyhedra is easy in the case of small cells. One can understand the structure of giant cells if one describes them in terms of packing of "polyhedra units." A "polyhedra unit" is constructed from two to five polyhedra.

## 2. Coordination Polyhedra

To define a coordination polyhedron, we first limit the coordination sphere. For this purpose Girgis (1) introduced the $\Sigma\left(r_{\mathrm{A}}+\right.$ $\left.r_{\mathrm{B}}\right) / d_{\mathrm{AB}}$ method. $\Sigma\left(r_{\mathrm{A}}+r_{\mathrm{B}}\right)$ is the sum of the radii of the atoms concerned (2) and $d_{\mathrm{AB}}$ is the interatomic distance between the central atom A and its neighbor B . The $\Sigma\left(r_{\mathrm{A}}+\right.$ $\left.r_{\mathrm{B}}\right) / d_{\mathrm{AB}}$ values versus the corresponding numbers of atoms are represented in a histogram. The widest range in which no atoms are present is called the "maximum gap" (3). $95 \%$ of the cases studied show such a "maximum gap" $(4,5)$. The number of nearest neighbors before this "maximum gap" is taken to be the conventional coordination number (CCN). The corresponding atoms (neighbors) constitute the coordination polyhedron around the atom considered (1). An example is shown in Fig. 1.


Fig. 1. Distribution of near neighbors as a function of their $\left(r_{\mathrm{A}}+r_{\mathrm{B}}\right) / d_{\mathrm{AB}}$ values for Cr in the $\mathrm{Cr}_{3} \mathrm{Si}$ structure type.

Table I shows the coordination numbers of the individual atoms in the treated structure types determined by the "maximum gap" method.

## 3. The Principle of Describing Structures with Coordination Polyhedra as Building Units

### 3.1. Procedure

To describe a structure using coordination polyhedra as building units one must first determine the coordination polyhedra of all the atomic positions of the structure by the 'maximum gap'" method mentioned above. The structure is described by packing the least number of polyhedra types (one or two are sufficient in most cases). If there is more than one possibility to describe the structure, certain priorities, which we mentioned in Girgis and Villars (8) are considered.

All the atoms in the unit cell must be included in the structure building polyhedra. The polyhedra should not interpenetrate.

### 3.2. Validity of This Model

The polyhedral packing principle has proven its validity for all structure types treated. More details are found in references ( $1,5-8$ ). Therefore it can be expected that this principle is valid for all in-

TABLE I
Coordination Numbers of the Individual Atoms in the Treated Structure Types

| Compound | Atom | Eq. pts. | CN |  |
| :---: | :---: | :---: | :---: | :---: |
| $\mathrm{Ag}_{8} \mathrm{Ca}_{3}$ | Ag 1 | 8 c | 12 |  |
|  | Ag 2 | 24h | 12 |  |
|  | Ca 1 | $12 e$ | 16 |  |
| $\mathrm{Cd}_{45} \mathrm{Sm}_{11}$ | Cd 1 | $4 b$ | 16 |  |
|  | Cd 2 | $4 d$ | 14 |  |
|  | Cd 3 | $16 e$ | 8 |  |
|  | Cd 4 | $16 e$ | 14 |  |
|  | Cd 5 | $16 e$ | 14 |  |
|  | Cd 6 | $16 e$ | 14 |  |
|  | Cd 7 | $24 f$ | 13 |  |
|  | Cd 8 | 24 g | 13 |  |
|  | Cd 9 | 48h | 12 |  |
|  | Cd 10 | $48 h$ | 15 |  |
|  | Cd 11 | $48 h$ | 12 |  |
|  | Cd 12 | $48 h$ | 12 |  |
|  | Cd 13 | $48 h$ | 13 |  |
|  | Sm 1 | $4 a$ | 14 |  |
|  | Sm 2 | 4 c | 16 |  |
|  | Sm 3 | $16 e$ | 16 |  |
|  | Sm 4 | $16 e$ | 16 |  |
|  | Sm 5 | 48h | 12 |  |
| $\mathrm{Cu}_{41} \mathrm{Sn}_{11}$ | Cu 1 | $16 e$ | 12 |  |
|  | Cu 2 | $16 e$ | 12 |  |
|  | Cu 3 | $16 e$ | 12 |  |
|  | Cu 4 | $16 e$ | 12 |  |
|  | Cu 5 | $16 e$ | 12 |  |
|  | Cu 6 | $16 e$ | 13 |  |
|  | Cu 7 | $16 e$ | 12 |  |
|  | Cu 8 | $24 f$ | 13 |  |
|  | Cu 9 | 24 g | 13 |  |
|  | Cu 10 | $24 f$ | 13 |  |
|  | Cu 11 | $48 h$ | 11 |  |
|  | Cu 12 | $48 h$ | 13 |  |
|  | Cu 13 | $48 h$ | 11 |  |
|  | Sn 1 | $16 e$ | 12 |  |
|  | Sn 2 | 24 g | 13 |  |
|  | Sn 3 | $48 h$ | 11 |  |
| $\mathrm{Mn}_{23} \mathbf{T h} \mathbf{h}_{6}$ | Mn 1 | $4 b$ | 14 |  |
|  | Mn 2 | 24d | 12 |  |
|  | Mn 3 | 32d | 13 |  |
|  | Mn 4 | 32 f | 12 |  |
|  | Th 1 | $24 e$ | 16 |  |
| $\mathrm{NiTi}_{2}$ | Ni 1 | $32 e$ | 12 |  |
|  | Ti 1 | $16 c$ | 12 |  |
|  | Ti 2 | $48 f$ | 14 | (center $3 m$ ) |
| $\mathbf{S b}_{\mathbf{2}} \mathbf{T l} \mathbf{7}_{7}$ | Sb 1 | $12 e$ | 14 |  |
|  | T1 1 | $2 a$ | 14 |  |
|  | Tl 2 | $16 f$ | 14 |  |
|  | Tl 3 | 24h | 14 |  |

termetallic structures. We studied more than 50 structure types out of 106 with more than 5 representatives (11). In the future, such models will be even more important because of the enormous amount of data available. In the new Pearson's Handbook (12) there are more than 2200 structure types.

## 4. Structure Description

Table II summarizes the general remarks and abbreviations used in this paper.

## 4.1. $\mathrm{Ca}_{3} \mathrm{Ag}_{8}$ Structure Type

$\mathrm{Ca}_{3} \mathrm{Ag}_{8}$ crystallizes in the cubic space group $\operatorname{Im} 3 m$ with 44 atoms per cell. The structure can be constructed from Agl icosahedra (Fig. 2a), symmetry $\overline{3} m-D_{3 d}$.

Four icosahedra at the same height, whose central atoms at $z=0.25$, constitute the lower polyhedral layer of the unit cell, sharing perpendicular triangular faces as shown in Fig. 2b. A mirror image polyhedra layer, whose central atoms at $z=0.75$, which lies over it, shares horizontal trian-


FIG. 2. Description of the $\mathrm{Ca}_{3} \mathrm{Ag}_{8}$ structure as a three-dimensional network of icosahedra connected by triangular faces. (a) Structure building polyhedron Agl. (b) The lower and the higher layer in the unit cell.

TABLE II
Summary of the General Remarks and Abbreviations Used in the Present Paper

| 州 | Linkage (contact) of the polyhedra of the same layer. |
| :---: | :---: |
| \% | Linkage (contact) between polyhedra of the first and second or the third and fourth layers, respectively. |
| \% | Linkage of the polyhedra between 'dou ble layers,'" i.e., between the second and the third layers ( $\mathrm{Ti}_{2} \mathrm{Ni}, \mathrm{Th}_{6} \mathrm{Mn}_{23}$, $\mathrm{Cu}_{41} \mathrm{Sn}_{11}, \mathrm{Sm}_{11} \mathrm{Cd}_{45}$ ) or connection to neighboring unit cells at different heights ( $\mathrm{Ca}_{3} \mathrm{Ag}_{8}, \mathrm{Tl}_{7} \mathrm{Sb}_{2}$ ). |
| \% | The central polyhedron in a schematic representation. |
| CN | Coordination number. |
| $\Delta$ | Triangular face. |
| $\square$ | Quadrangular face. |

gular faces. The connection to the neighboring unit cells is the same as mentioned above. Each icosahedron shares a triangular face with 6 neighboring icosahedra, namely two in the $x, y$, and $z$ directions, respectively.

## 4.2. $\mathrm{Tl}_{7} \mathrm{Sb}_{2}$ Structure Type

The $\mathrm{Tl}_{7} \mathrm{Sb}_{2}$ structure crystallizes in the cubic space group $\operatorname{Im} 3 m$ with 54 atoms per cell. The structure can be constructed from Tl 1 and Tl 2 polyhedra. Tl 1 is a tetrakishexahedron ( $24 \triangle$; symmetry $m 3 m-O_{h}$ ) and Tl2 is a distorted rhombic dodecahedron (symmetry $3 m-C_{3 v}$ ) Fig. 3a. A Tl1 polyhedron at the center of the unit cell shares an atom with four Tl2 polyhedra; namely the two Tl2 polyhedra along [110] share an atom at $z=0.33$ with the central Tl1 polyhedron while the two along [110] (mirror images of those along [110]), share an atom at $z=0.67$ with it (Fig. 3c, schematic representation Fig. 3b).

The connections to the neighboring cells are perpendicular quadrangular faces between Tl2 polyhedra shared along the cell
diagonals, constituting "double polyhedra." The connection in the $z$ direction occurs between T12 double polyhedra of the [110] and [1T0] direction belonging to unit cells at different heights, sharing quadrangular faces (Fig. 3d). Thus each Tl2 polyhedron shares a quadrangular face with three other Tl2 polyhedra and with one Tl1 polyhedron an atom.

### 4.3. Ti ${ }_{2} \mathrm{Ni}$ Structure Type

$\mathrm{Ti}_{2} \mathrm{Ni}$ crystallizes in the cubic space group $F d 3 m$ (origin at 3 m ) with 96 atoms per cell. The structure can be constructed from Til polyhedra (icosahedra); symmetry $3 m-D_{3 d}$ (Fig. 4a).

There are four layers of chains in the unit cell, the heights of the central atoms are $0.0,0.25,0.5,0.75$. Each chain consists of icosahedra sharing perpendicular triangular faces. Chains of the same layer have no contact. The chains of the four different layers are superimposed in the [110] and the [1T0] direction alternatively and share a triangular face with two icosahedra of the layer above as well as below (Fig. 4c).

The contact to neighboring cells is the same as described above (Fig. 4b). Thus each icosahedron shares a triangular face with six further icosahedra.

## 4.4. $\mathrm{Th}_{6} \mathrm{Mn}_{23}$ Structure Type

$\mathrm{Th}_{6} \mathrm{Mn}_{23}$ crystallizes in the cubic space group $\mathrm{Fm} 3 m$ with 116 atoms per cell. The structure can be constructed from Mn3 polyhedra consisting of three quadrangular and 16 triangular faces; symmetry $3 m-C_{3 v}$ (Fig. 5a).
$\mathrm{Mn}_{3}$ polyhedra at the same height (central atom: $z=0.12$ ) are interconnected by perpendicular triangular and quadrangular faces alternatively, forming chains along the [110] direction, which have no contact among themselves. A second layer of identical chains (central atom: $z=0.38$ ) which is a mirror image of the first one, lies between the chains of the first layer sharing


Fig. 3. Description of the $\mathrm{Tl}_{7} \mathrm{Sb}_{2}$ structure as a three-dimensional network of T 12 "double polyhedra" connected by T11 polyhedra. (a) Structure building polyhedra T11 and T12. (b) Schematic representation of the five-polyhedron unit which corresponds to the unit cell. (c) The unit cell contents and two T12 polyhedra of the neighboring cell at the same height. (d) Linkage of a Tl2 "double polyhedron" with Tl2 polyhedra of the lower cells. (e) Schematic representation of the polyhedra packing. The borders of the unit cell are shown as well as the height range of the coordination atoms in the polyhedra.


Fig. 4. Description of the $\mathrm{Ti}_{2} \mathrm{Ni}$ structure as a three-dimensional network of horizontal chains of Ti icosahedra (sequences of layers alternatively [110], [1T0], [110], [110]). (a) Structure building polyhedron Ti1. (b) Linkage of a polyhedra chain with the upper and lower one. (c) First and second; third and fourth layer in the unit cell.
atoms at $z=0.25$ (Fig. Sc). An identical "double layer"' of chains, but running along [110] and elevated by $z=0.5$, shares quadrangular faces with the lower one; (Figs. 5b and c).

The connections to the neighboring cells correspond to those within the cell. Thus each polyhedron shares an atom, a triangular or a quadrangular face with six other polyhedra.


Fig. 5. Description of the $\mathrm{Th}_{6} \mathrm{Mn}_{23}$ structure as a three-dimensional network of horizontal chains of Mn3 polyhedra (sequence of layers: [110], [110], [110], [1T0]. (a) Structure building polyhedron Mn3. (b) Linkage of a chain with the upper and lower one. (c) Upper and lower "double layer" in the unit cell.

## 4.5. $\mathrm{Cu}_{41} \mathrm{Sn}_{11}$ Structure Type

$\mathrm{Cu}_{41} \mathrm{Sn}_{11}$ crystallizes in the cubic space group $F 43 m$ with 416 atoms per cell. The structure can be constructed from Cull, $\mathrm{Cu} 12, \mathrm{Cu} 13$, and Sn 3 polyhedra. Cu11, Cu 13 , and $\operatorname{Sn} 3$ polyhedra are similar and consist of four quadrangular and 10 triangu-
lar faces; symmetry $m-C_{s}$; the Cu12 polyhedron consists of two quadrangular and 18 triangular faces; symmetry $m-C_{s}$ (Fig. 6a).

We first consider the lowest layer (Fig. 6c below, bold lines) and distinguish four quadrants consisting of four polyhedra (four-polyhedra unit) each. Within the units the polyhedra share atoms; the two polyhe-

dra along [ $1 \overline{10} 0$ ] are mirror images of those along [110]. The first and the fourth quadrant are occupied by Sn 3 , the second and the third by Cu13 polyhedra. The different units of four polyhedra share quadrangular faces while the units of the same kind share edges along [110] and [110] directions. The second layer (Fig. 6c below, thin lines) consists of Cu11 and Cu12 polyhedra units, arranged as the first layer, but shifted by about a quarter of the cell diagonal. Thus each quadrant contains Cull as well as Cu 12 polyhedra (Fig. 6b shows half a quadrant of each layer).

There are four layers in the cell, consisting alternatively of $\mathrm{Sn} 3-\mathrm{Cu} 13$ and $\mathrm{Cul1-}$ Cu12 units; each layer is displaced by about a quarter of the cell diagonal. In the $z$ direction the polyhedra share corners and triangular faces (Fig. 6b). The schematic arrangement of the polyhedra in the unit cell is represented in Fig. 6d.

As connections to the neighboring cells correspond to those within the unit cell, each polyhedron is connected to eight other polyhedra sharing corners, edges, triangular (only between Cu12 and Cu13 polyhedra ) and quadrangular faces.

## 4.6. $S m_{11} C d_{45}$ Structure Type

$\mathrm{Sm}_{11} \mathrm{Cd}_{45}$ crystallizes in the cubic space group $F 43 m$ with 448 atoms per cell. The structure can be described by packing Cd2 ( $24 \triangle$ faces; symmetry $\overline{4} 3 m-T_{d}$ ), Cd4 (18 $\triangle$ and $3 \square$ faces; symmetry $3 m-C_{3 v}$ ), Sm2 ( $28 \Delta$ faces; symmetry $43 m-T_{d}$ ) and $\operatorname{Sm} 3$ ( $28 \triangle$ faces; symmetry $3 m-C_{3 v}$ ) polyhedra. Both Cd polyhedra are highly distorted tetrakishexahedra while Sm polyhedra are distorted Kasper-16 polyhedra (Fig. 7a).

The packing principle corresponds to that of the $\mathrm{Tl}_{7} \mathrm{Sb}_{2}$ type (Figs. 3b and 7b).

Each Cd 2 and Sm 2 polyhedron is surrounded by four Cd 4 or four Sm 3 polyhedra, respectively (Fig. 7c). In the lower Cd unit (Fig. 7d left, thin lines) the Cd4 polyhedra along [110] share an atom at $z=0.33$ with the central Cd 2 polyhedron while those along [110], mirror images of those along [110], share an atom at $z=0.17$ with it. In the Sm unit, the polyhedra share triangular faces instead of atoms (Fig. 7d, right).

Four units at the same height (central atoms Cd 2 and Sm 2 at $z=0.25$ ) constitute the lower layer (Fig. 7e, right) whereas the first and the fourth quadrant are occupied by Sm units sharing a triangular face; the second and the third are occupied by Cd units sharing a quadrangular face. Sm and Cd units have no contact.

The unit cell has two layers; the upper layer is shifted by $0 \frac{1}{2} \frac{1}{2}$ so that Cd units lie above Sm units and vice versa (Fig. 7e), but without any connection between them. Cd4 or Sm3 'double polyhedra'" along [110] of the upper layer overlap with similar ones along [110] of the lower layer sharing triangular faces for the Sm and quadrangular faces for the Cd units (Figs. 7d, e).

As the connections to the neighboring cells correspond to those within the cell, each polyhedron shares atoms and quadrangular faces (Cd units) or triangular faces only ( Sm units) with four other polyhedra.

## 5. Conclusions and Discussion

1. The polyhedral packing principle has proved valid for describing all structures treated (Table I). We studied earlier more than 50 structure types. It can therefore be expected that this principle is valid for all intermetallic structure types.

Other structure describing models are

[^0]

(c)


(d)


Fig. 7. Description of the $\mathrm{Sm}_{11} \mathrm{Cd}_{45}$ structure as a three-dimensional network of Cd 2 and Cd 4 polyhedra interpenetrating another three-dimensional network of Sm 2 and Sm 3 polyhedra without any contact. (a) Structure building polyhedra Cd2, Cd4, Sm2, and Sm3. (b) Schematic representation of a five-polyhedra-unit, the "building unit" of the structure. (c) Cd and Sm five-polyhedra building unit. (d) Linkages in the Cd and Sm network, respectively. (e) Schematic representation of the polyhedrapacking in the unit cell; Sm polyhedra as circles, Cd polyhedra as squares; Cd2 and Sm2 polyhedra are hatched.


Fig. 7-Continued.
valid only for certain structures. But the validity for all structures has not been studied in general. Samson's (9) fascinating model for giant molecules, for example, describes the structure by packing Friauf polyhedra. This model is therefore limited to structures the atoms of which are surrounded by Friauf polyhedra. Another model is describing structures by stacking "Kagomé nets." This is again limited to structures which exhibit layers of atoms constituting Kagomé nets.
2. The polyhedral packing principle delivers a simple representation of the structures; e.g., $\mathrm{Ca}_{3} \mathrm{Ag}_{8}$ (cI44) can be described by packing icosahedra. The representation of structures with hundreds of atoms per cell proved more complicated. Even then the packing can be understood if one knows the packing unit and its contacts to the surrounding units, e.g., four-polyhedra-units for the $\mathrm{Cu}_{41} \mathrm{Sn}_{11}$ and five-polyhedra-units for the $\mathrm{Sm}_{11} \mathrm{Cd}_{45}$ structure type.
3. Icosahedra, 14 and 16 Kasper polyhedra, tetrakishexahedra, and rhombic dodecahedra play an important part in the de-
scription of these structures, although they belong to different structure types. In general cubes, cubo-octahedra, bicapped square antiprism beside those mentioned above are important in the description of intermetallic structures. This demonstrates their tendency to have high symmetry and coordination numbers.
4. The mean coordination number of the treated structures is equal to 13.2 which reinforces Laves' (10) statement that the metals and intermetallic phases tend to have high coordination numbers.
5. The small number of cubic structures described in this paper shows that certain "building blocks" repeatedly occur; namely chains along the cell diagonals in $\mathrm{Th}_{6} \mathrm{Mn}_{23}$ and $\mathrm{Ti}_{2} \mathrm{Ni}$ structures; the same five-polyhedra units in $\mathrm{Tl}_{7} \mathrm{Sb}_{2}$ and $\mathrm{Sm}_{11} \mathrm{Cd}_{45}$ whereas in $\mathrm{Tl}_{7} \mathrm{Sb}_{2}$ the unit consists of the whole unit cell content, in $\mathrm{Sm}_{11} \mathrm{Cd}_{45}$ just a part of it.

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[^0]:    Fig. 6. Description of the $\mathrm{Cu}_{41} \mathrm{Sn}_{11}$ structure as a network of two alternating layers consisting of two different polyhedra units each. (a) Structure building polyhedra $\mathrm{Cu} 11, \mathrm{Cu} 12, \mathrm{Cu} 13$ and Sn 3 . (b) Linkage of the layers in the $z$ direction. (c) First and second; third and fourth polyhedra layer in the unit cell. (d) Schematic representation of the polyhedra packing in the unit cell.

